

#### **LECTURE 16**

Dr. Teresa D. Golden University of North Texas Department of Chemistry

#### XRD can be used for:

- -Bravais lattice determination phase determination (crystalline phases and orientation)
- -Lattice parameter determination
- -Determination of solvus line in phase diagrams (order-disorder transformation)
- -Long range order
- -Crystallite size and Strain
- -Temperature factor thermal diffuse scattering (thermal expansion)

Advanced

-Thickness measurements of thin films and multilayers

#### XRD can be used for:

- -Bravais lattice determination phase determination (crystalline phases and orientation)
- -Lattice parameter determination
- -Determination of solvus line in phase diagrams (order-disorder transformation)
- -Long range order
- -Crystallite size and Strain
- -Temperature factor thermal diffuse scattering (thermal expansion)

Advanced

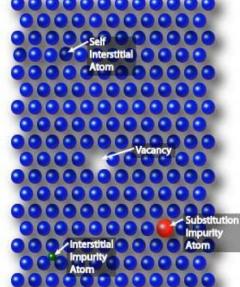
-Thickness measurements of thin films and multilayers

# Vectors and Planes Crystal Defects

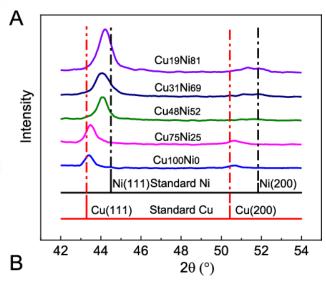
Point defects – substitutional or interstitial impurities.

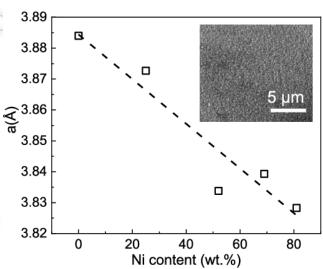
Vegard's law – volume of unit cells in a substitutional solid solution is linearly proportional to the fraction of sites

substituted.









#### Vegard's law

In crystallography, Vegard's law is an empirical finding resembling the rule of mixtures. In 1921, Lars Vegard discovered that the lattice parameter of a solid solution of two constituents is approximately a weighted mean of the two constituents' lattice parameters at the same temperature.



#### Phase Analysis - Vegard's Law

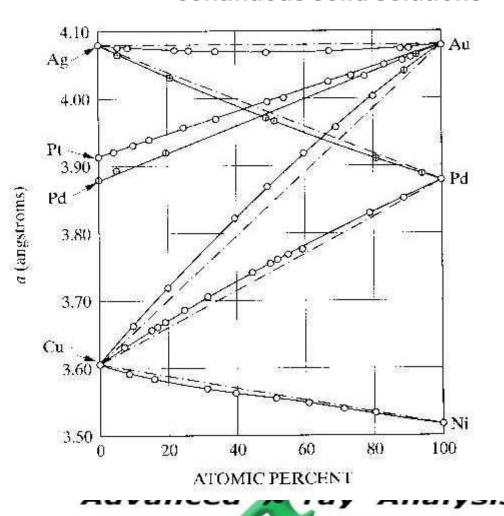
$$d_{hkl} = \sqrt{\frac{a^2}{h^2 + k^2 + l^2}}$$

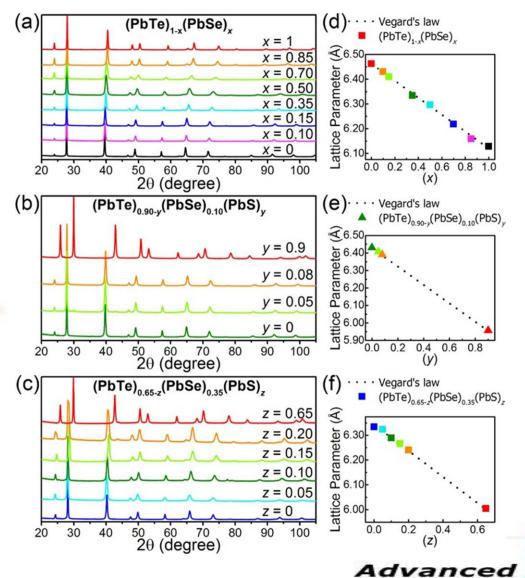
#### Ex) Au-Pd

- · To create the plot on the right
  - Using the crystal structure of the alloy calculate "a" for each metal
  - Draw a straight line between them as shown on the chart to the left.
- To calculate the composition
  - · Calculate "a" from d-spacings
  - "a" will be an atomic weighted fraction of "a" of the two metal

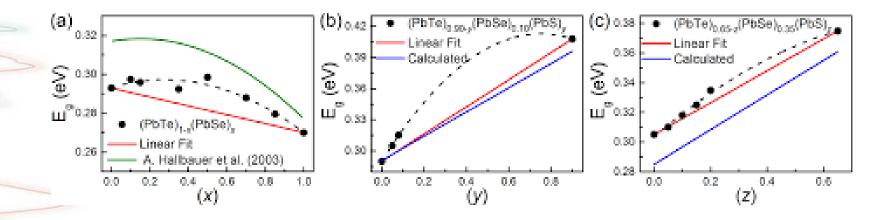
Figure 11-6 Lattice parameters of some continuous solid solutions. Dot-dash lines indicate Vegard's law. Barrett [1.7].

#### Good for alloys with continuous solid solutions





X-ray Analysis



#### **Measuring Changes In Phase Fraction**

**Using I/Icor** 

$$\frac{I_{\alpha-\exp}(hkl)}{I_{\beta-\exp}(HKL)} = \frac{I_{\alpha}(hkl) \times I_{lcor_{\beta}}(hkl) \times w_{\beta}}{I_{lcor_{\alpha}}(hkl) \times I_{\beta}(hkl) \times w_{\alpha}}$$

#### -Where

- $\bullet \quad \frac{I}{I_{cor}} = \frac{Intensity of \ sample's \ 100\% \ peak}{Intensity of \ Corundum' \ s \ 100\% \ peak}$
- ω= weight fraction
- I(hkl)=Reference's relative intensity
- I<sub>exp</sub>(hkl)=Experimental integrated intensity



#### **Measuring Changes In Phase Fraction**

**Using Direct Comparison Method** 

$$\frac{I_{\alpha-\exp}(hkl)}{I_{\beta-\exp}(HKL)} = \frac{R_{\alpha} \times v_{\alpha}}{R_{\beta} \times v_{\beta}}$$

$$R = \frac{1}{V} \left[ pCL_{P} \left[ F_{hkl} \right]^{2} \right]$$

$$I_{hkl} = I_{0}pCL_{P} \left[ F_{hkl} \right]^{2}$$

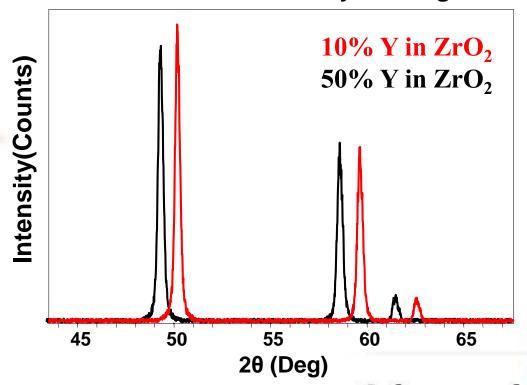
Because this is already a complicated method, many choose to go ahead and use Rietveld Refinement



#### Measuring Changes In Phase Fraction

Substitutional doping can change bond distances, reflected by a change in unit cell lattice parameters

The change in peak intensity due to substitution of atoms with similar Z is much more subtle and may be insignificant



$$R(Y^{3+}) = 0.104\text{Å}$$
  
 $R(Zr^{4+}) = 0.079\text{Å}$ 

Advanced X-ray Analysis

#### XRD can be used for:

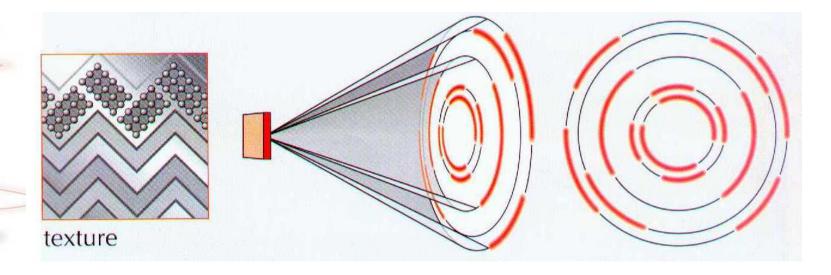
- -Bravais lattice determination phase determination (crystalline phases and orientation)
- -Lattice parameter determination
- -Determination of solvus line in phase diagrams (order-disorder transformation)
- -Long range order (Texture Analysis)
- -Crystallite size and Strain
- -Temperature factor thermal diffuse scattering (thermal expansion)

Advanced

-Thickness measurements of thin films and multilayers

# **Diffraction Theory**

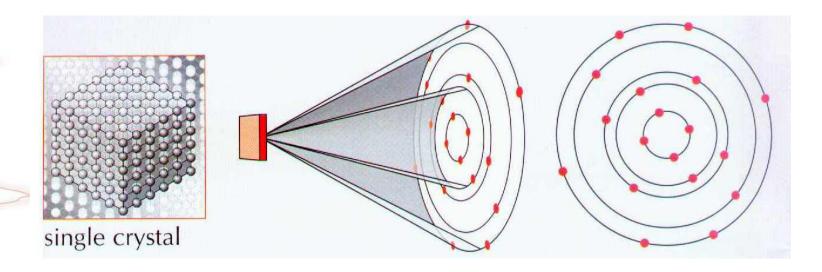
#### **Texture analysis**



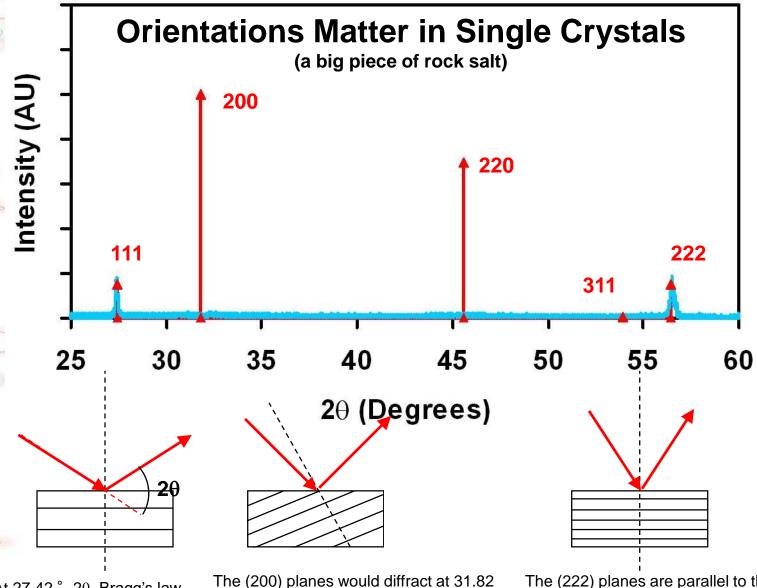


# **Diffraction Theory**

#### **Single Crystal**







At 27.42 ° 2θ, Bragg's law fulfilled for the (111) planes, producing a diffraction peak.

The (200) planes would diffract at 31.82 ° 2θ; however, they are not properly aligned to produce a diffraction peak

The (222) planes are parallel to the (111) planes.

#### **Texture**

Grains within a polycrystalline are not completely randomly distributed

There can be clustering of grains about some particular orientation(s) to a certain degree

Inducing texture in powder samples - causes peak intensities to vary

Large crystallite sizes and non-random crystallite orientations both lead to peak intensity variation

The measured diffraction pattern will not agree with that expected from an ideal powder

Advanced X

Texture in samples is a common occurrence, and usually have more to do with how the samples are made.

- -Plastically deformed metals (cold rolled, etc)
- -Powders with particle shapes related to their crystal structure, particular planes form the faces
- -Orientation in particular directions (Plates, needles, acicular, cubes, etc.)



Orientation in particular directions (Plates, needles, acicular, cubes, etc.)

-Fiber Texture

A particular direction [uvw] for all grains is more or less parallel to the wire or fiber axis

e.g. [111] fiber texture in Al cold drawn wire

Double axis is also possible

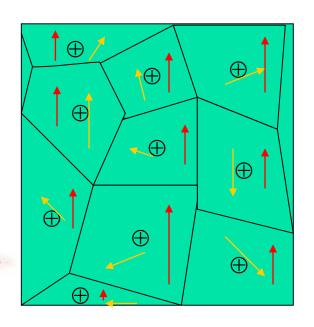
Example: [111] and [100] fiber textures in Cu wire

-Sheet Texture

Most of the grains are oriented with a certain crystallographic plane (hkl) roughly parallel to the sheet surface and certain direction [uvw] parallel to the rolling direction

Notation: (hkl)[uvw]





⊕ [uvw] i.e. perpendicular to the surface of all grains is parallel to a direction [uvw] Also, if the direction [u<sub>1</sub>v<sub>1</sub>w<sub>1</sub>] is parallel for all regions, the structure is like a single crystal

However, the direction  $[u_1v_1w_1]$  is not aligned for all regions, the structure is like a mosaic structure, also called as **Mosaic Texture** 

Advanced M-ray Analysis

Orientation in particular directions (Plates, needles, acicular, cubes, etc.)

**Simple Means of Quantifying Texture** 

Lotgering degree of orientation (f)

A comparison of the relative intensities of a particular family of (hkl) reflections to all observed reflections in a coupled  $2\theta$  powder x-ray diffraction (XRD) Spectrum f is specifically considered a measure of the "degree of

orientation" and ranges from 0% to 100%

po is p of a sample with a random crystallographic orientation.

$$f = \frac{p - p_o}{1 - p_o};$$

$$p = \frac{\sum I_{00l}}{\sum I_{00l} + \sum I_{\text{non-00}l}}$$



#### **Relative Texture Coefficient (RTC)**

Relative peak intensities are analyzed by determining relative texture coefficients of the reflections present in the XRD pattern.

$$RTC_{hkl} = \frac{I_{p,hkl}/I^{\circ}_{p,hkl}}{\sum I_{p,hkl}/I^{\circ}_{p,hkl}} *100\%$$

where  $I_{p,hkl}$  is the intensity of the peak for each sample at each (hkl) reflection used.

 $I^{\circ}_{p,hkl}$  is the intensity of those reflections for the standard reference (PDF# 00-000-0000) in the database.

The standard provides a random orientation pattern to compare against any preferred orientation found in the samples.



Relative Texture Coefficient (RTC) Example: Golden et.al.

X-ray diffraction (XRD)
of (a) pure Ni, (b) Ni CeO<sub>2</sub>, and
(c) Ni Fc-CeO<sub>2</sub> coatings

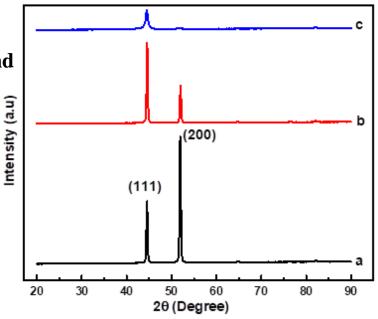
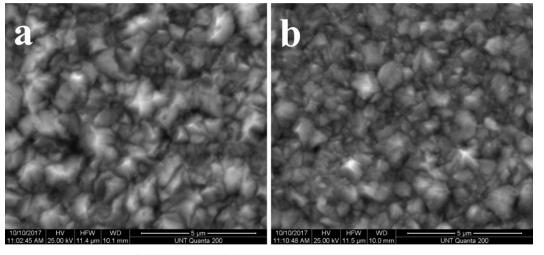
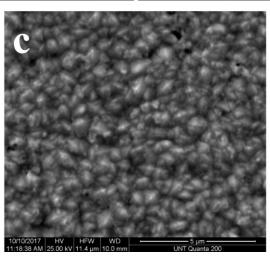


Table. XRD reflection intensities, relative texture coefficients, and crystallite sizes for Nickel, Ni CeO<sub>2</sub> and Ni Fc-CeO<sub>2</sub> electrodeposited coatings.

	Coating (n=3)	<b>I</b> ( <b>f</b> ) <sub>111</sub>	I(f) <sub>200</sub>	I(f) <sub>220</sub>	RTC <sub>111</sub>	RTC <sub>200</sub>	RTC <sub>220</sub>	Crystallite size (nm)
	PDF# 00-004-0850	100	42	21				
	Nickel	49	100	3	29.8	63.3	6.9	$47.6 \pm 5.0$
7	Ni CeO <sub>2</sub>	100	51	4	50.7	41.9	6.7	$43.3 \pm 6.3$
	Ni Fc-CeO <sub>2</sub>	100	11	16	63.1	14.6	22.3	$32.0 \pm 5.5$

Relative Texture Coefficient (RTC) Example: Golden et.al.







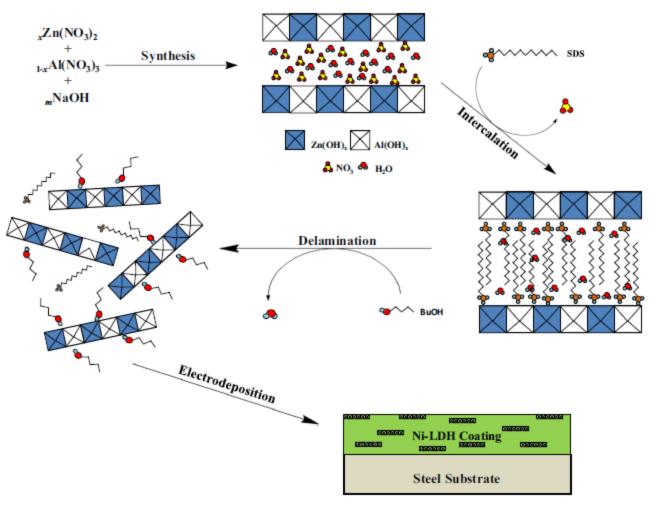


Fig. 1. Overview of the layered double hydroxide colloid synthesis and Ni-LDH metal matrix nanocomposite coating process.



Table 2

Crystallite size and strain measured from Williamson-Hall analysis and results of relative texture coefficient analysis of the XRD patterns of Ni and Ni-LDH coatings.

	Crystallite size (nm)	Strain	RTC <sub>111</sub>	RTC200	RTC <sub>220</sub>	RTC <sub>311</sub>
В1	37.54 ± 27	$0.0228 \pm 0.013$	10.04 ± 5.5	5.84 ± 2.2	75.93 ± 9.0	8.19 ± 1.5
B2	60.20 ± 21	$0.0192 \pm 0.001$	9.86 ± 6.0	8.66 ± 4.6	72.36 ± 12.0	$9.12 \pm 2.0$
BL1	$41.16 \pm 2$	$0.0120 \pm 0.007$	$6.25 \pm 0.2$	$3.40 \pm 1.0$	84.94 ± 4.5	$5.41 \pm 3.3$
BL2	25.10 ± 6	$0.0158 \pm 0.017$	$8.22 \pm 1.8$	5.88 ± 3.3	78.91 ± 8.6	6.99 ± 3.7

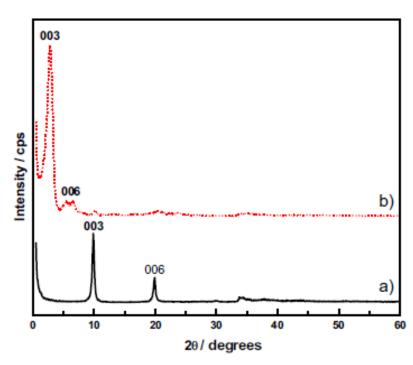


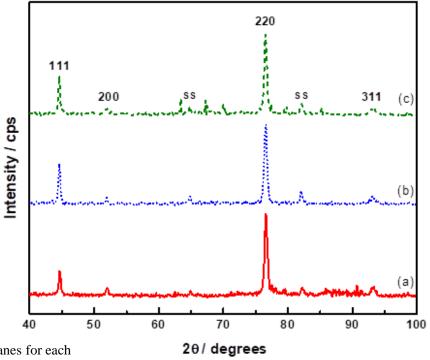
Fig. 3. Powder XRD patterns of a) LDH-NO<sub>3</sub> (solid line), and b) LDH-DS (dotted line) powders.



Relative Texture Coefficient (RTC) Example: Golden et.al.

$$RTC_{hkl} = \frac{I_{p,hkl}/I^{\circ}_{p,hkl}}{\sum I_{p,hkl}/I^{\circ}_{p,hkl}} \times 100\%$$

where *I*p,hkl is the intensity of the peak for each sample at (111), (200), (220), and (311) reflections and *I*°p,hkl is the intensity of those reflections for the nickel standard reference (PDF# 00-004-0850)



**Table 2:** Relative texture coefficients for (111), (200), (220), and (311) growth planes for each sample as measured with XRD

sample as	s illeasured with An	.D.			
Sample	RTC <sub>111</sub>	RTC200	RTC220	RTC311	
<b>A1</b>	5.26	4.85	72.32	17.66	
B1	8.09	4.32	71.84	15.74	
<b>B2</b>	7.80	4.27	72.66	15.26	



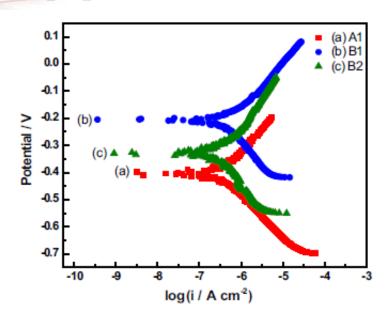


Fig. 7 Tafel polarization curves for each coating a A1 (squares), b B1 (circles), and c B2 (triangles) run in 3.5% NaCl after 14 days immersion in 3.5% NaCl

Table 2 Relative texture coefficients for (111), (200), (220), and (311) growth planes for each sample as measured with XRD

Sample	RTC <sub>111</sub>	RTC <sub>200</sub>	RTC <sub>220</sub>	RTC <sub>311</sub>
A1	5.26	4.85	72.32	17.66
B1	8.09	4.32	71.84	15.74
B2	7.80	4.27	72.66	15.26

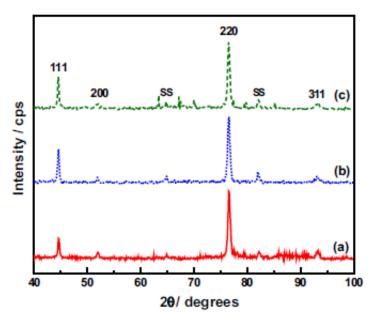


Fig. 2 X-ray diffraction patterns of each nickel coating a A1 (solid line), b B1 (dotted line), and c B2 (dashed line)

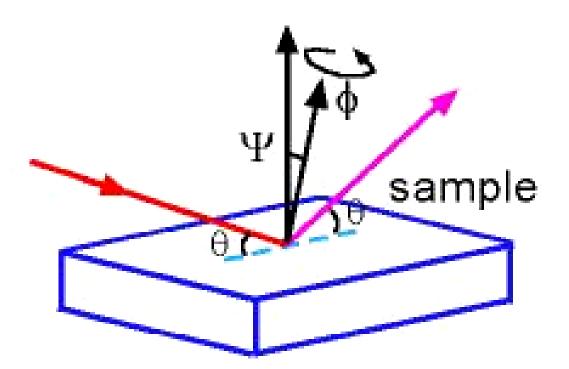
#### **Pole Figure Analysis**

**Texture Analysis Experiment** 

Measurement of texture (the non-random or preferred orientation of crystallites) involves measurement of the variations in intensity of a single Bragg reflection as the sample is both tilted (psi) and rotated (phi).

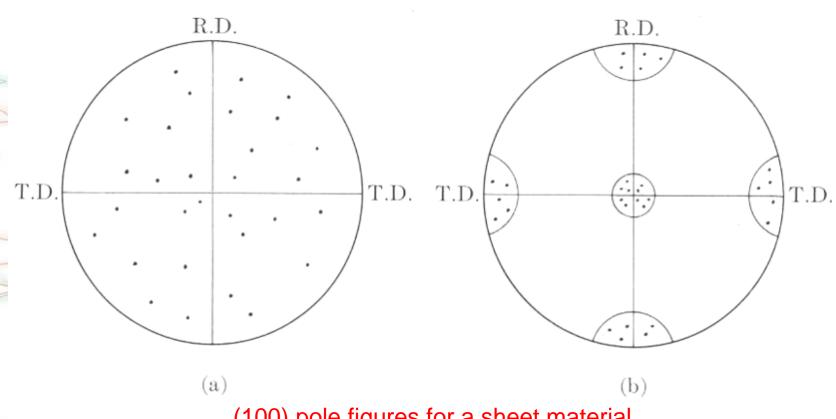
The result is plotted as a 'pole figure', in which the contours indicate intensity levels as a function of sample orientation.





Pole Figure Measurement

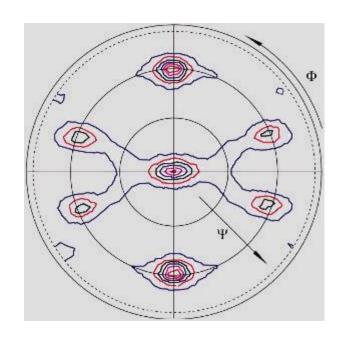




(100) pole figures for a sheet material(a) Random orientation (b) Preferred orientation

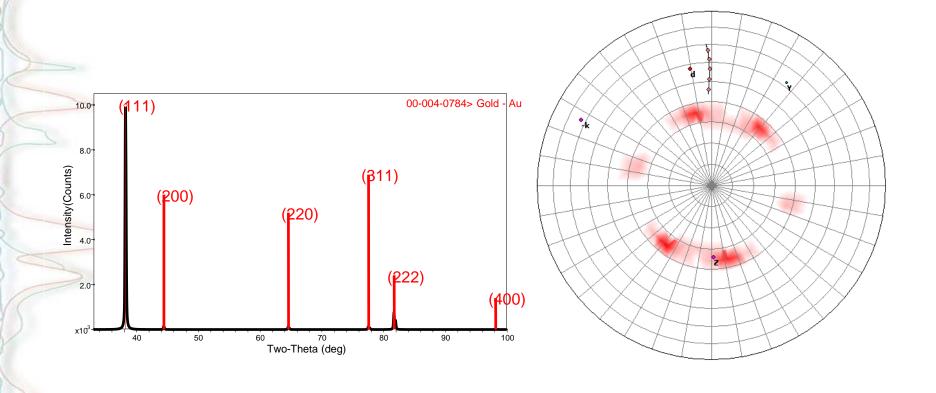
Advanced X-ray Analysis

#### **Texture Analysis**



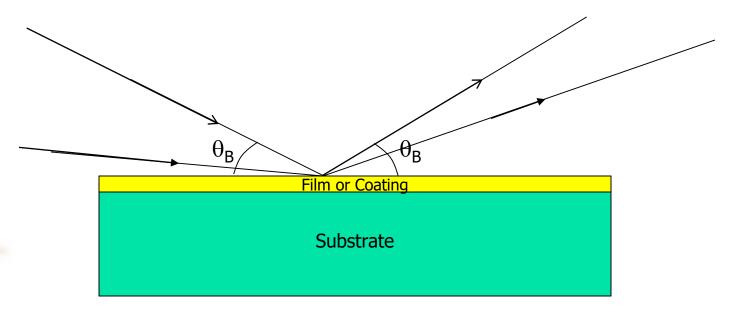


#### **Texture Analysis**





#### **Thin Film Analysis**



Example – thin films grown epitaxially on top of a substrate.



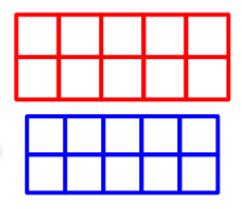
#### **Epitaxial Films**

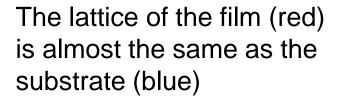
- An epitaxial film is a lattice-matched thin film grown on usually a single crystal substrate
- The lattice of the film is nearly identical to the lattice of the substrate
- When the film grows, its lattice changes (strains) in order to match the lattice of the substrate
- The atomic bonding across the substrate and film is "perfectly" matched

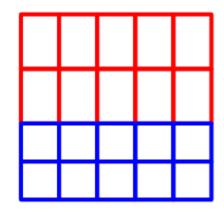


**Epitaxial Films** 

An epitaxial film is a lattice-matched thin film grown on usually a single crystal substrate







The lattice of the epitaxial film (red) distorts to minimize the strain energy where it bonds to the substrate (blue)



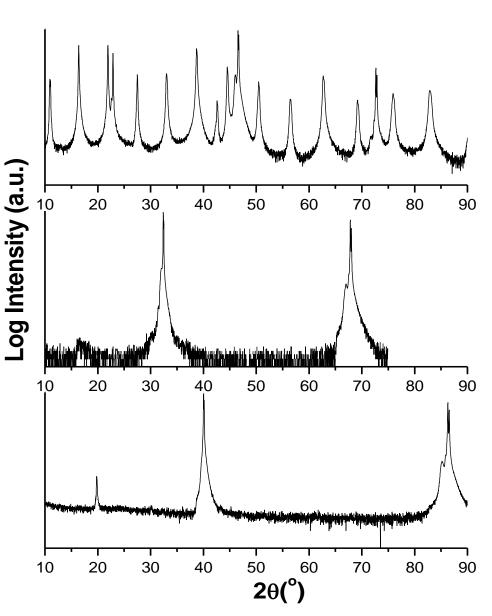
#### **Example:**

Bismuth Titanate thin films on oriented SrTiO<sub>3</sub> substrates

Only one type of peaks

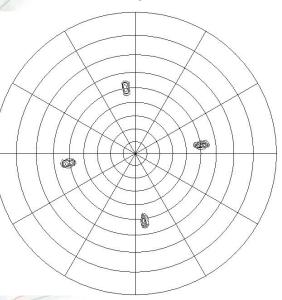
It apparent that films are highly oriented

3 different examples



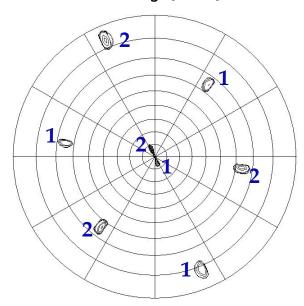
#### **Pole Figure Analysis**

SrTiO<sub>3</sub> (100)



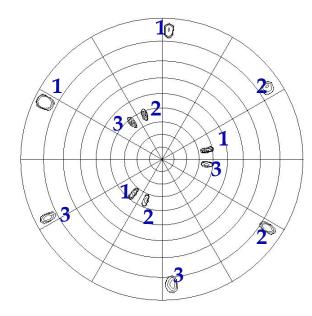
- 4 Peaks at ~50°
- Excellent in-plane orientation

SrTiO<sub>3</sub> (110)



- 2 sets of peaks at ~ 5, 65 and 85°
- Indicating a doublet or opposite twin growth

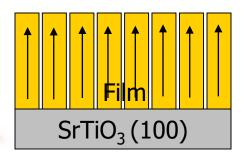
SrTiO<sub>3</sub> (111)

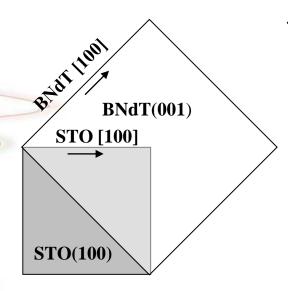


- 3 sets of peaks at ~ 35 and 85°
- indicating a triplet or triple twin growth

Advanced X-ray Analysis

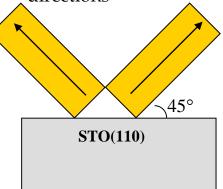
#### **Texture Analysis Evolution**





BNdT/SrTiO3 (100)

Two (100) planes inclined at 45° to (110) plane in opposite directions



3 (100) planes inclined at 54.7° to (110) plane, separated by 120°

STO(111)

54.7°

STO(111)

BNdT/SrTiO<sub>3</sub> (110)

BNdT/SrTiO<sub>3</sub> (111)



#### **Rocking Curves**

A useful method for evaluating the quality of oriented samples such as epitaxial films.

RC measurement reveals broadening of the diffraction peaks. The increase in the peak widths can be caused by:

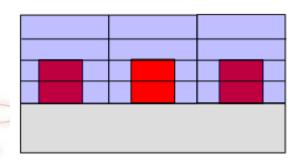
- -Mosaicity (misorientation of crystallites)
- -Strain
- -Limited layer thickness



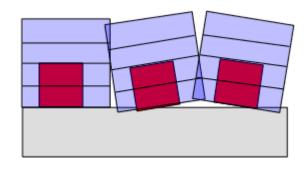
**Epitaxial Films – Mosaic Spread** 

Mosaicity is created by slight misorientations of different crystals as they nucleate and grow on the substrate.

When the crystals join, they form low energy domain boundaries.



Ideal case, each nuclei (red) is perfectly oriented. When the crystals grow and meet, there is perfect bounding between the crystallites and therefore there is no grain boundary



If the nuclei (red) are slightly misaligned, then low angle domain boundaries will be formed.



#### **Rocking Curves**

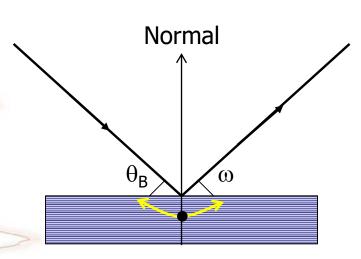
Typically XRD peak width is quantitatively described by Full-Width-at Half-Maximum (FWHM).

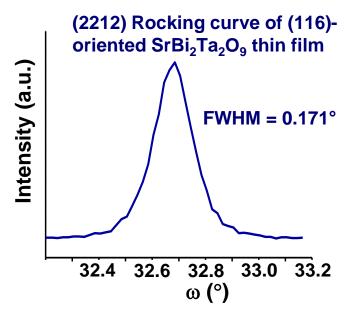
 $\omega$  is changed by rocking the sample but  $\theta_B$  is held constant

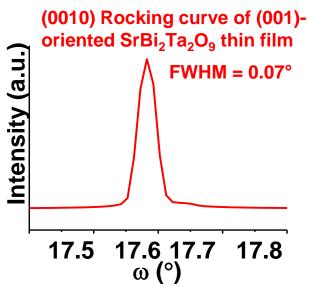
Width of Rocking curve is a direct measure of the range of orientation present in the irradiated area of the crystal



#### **Rocking Curves**







#### Summary

٠.									
	Perfect Epitaxy	Single crystal film in perfect registry with a substrate. There are no defects in the film or the substrate.							
11 × 1/	Nearly perfect epitaxy	Single crystal film in nearly perfect registry with a substrate. Both film and substrate contain a low concentration of defects. Most defects are dislocations in the film.							
	Textured epitaxial*	Film consists of mosaic domains in nearly perfect registry with the substrate. All domain boundaries are very low angle/low energy. There is nearly perfect bonding across domain boundaries.							
	Strongly textured polycrystalline	Film consists of grains with nearly perfect preferential orientation of all principle axes. This orientation is often strongly correlated to the substrate. Misorientation parameter for texture is small.							
	Textured polycrystalline	Film consists of grains with a preferred orientation for 3 principle axes or only along 1 axis out-of-plane.							
)	Polycrystalline	Film consists of randomly oriented grains.							
	Amorphous	Film does not have long-range crystalline order.							

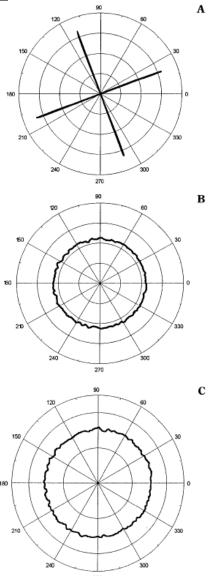


#### Summary

	Thickness	Composition	Lattice Strain/ Relaxation	Defects	Orientation	Residual Stress	Crystallite Size
Perfect Epitaxy	XRR, HRXRD	HRXRD, RC	Assume 100%	Assume none	HRXRD		
Nearly perfect epitaxy	XRR, HRXRD	HRXRD, RC	HRXRD	RC	HRXRD		
Textured epitaxial*	XRR, HRXRD	HRXRD	HRXRD, IP- GIXD	RC	HRXRD		
Strongly textured polycrystalline	XRR	XRPD, IP- GIXD	IP-GIXD	XRPD, IP-GIXD	IP-GIXD, PF	IP-GIXD	XRPD, IP- GIXD
Textured polycrystalline	XRR	XRPD, GIXD or IP- GIXD		XRPD, GIXD OR IP-GIXD	PF	Psi	XRPD, GIXD
Polycrystalline	XRR	XRPD, GIXD		XRPD, GIXD	PF	Psi	XRPD, GIXD
Amorphous	XRR						

XRR- X-Ray Reflectivity HRXRD- High Resolution XRD using coupled scan or RSM RC- Rocking Curve
XRPD- Bragg-Brentano powder diffraction
GIXD- grazing incidence XRD

IP-GIXD- in-plane grazing incidence XRD
PF- pole figure
Psi- sin<sup>2</sup>psi using parellel beam



**Figure 7.** Rotational scans for (A) [100] silicon single crystal, (B) silicon powder, and (C) a 5  $\mu$ m thick [100] Tl<sub>2</sub>O<sub>3</sub> film. X-ray diffraction is from the {111} planes.

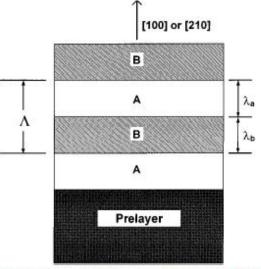


Figure 8. Idealized superlattice grown on an oriented prelayer. The orientation of the prelayer is maintained through the superlattice. The thickness of layer a is  $\lambda_a$ , and the thickness of layer b is  $\lambda_b$ .

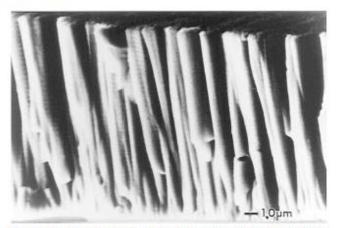


Figure 9. Scanning electron micrograph of an electrodeposited columnar superlattice structure. The individual layer thicknesses are approximately  $0.5 \, \mu \text{m}$ . The bar marker is  $1.0 \, \mu \text{m}$ .

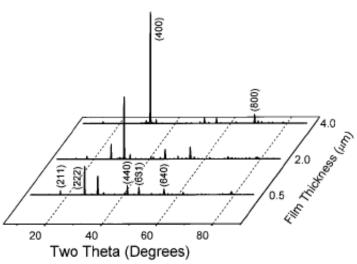


Figure 3. Dependence of the prelayer texture on the film thickness for a  $Tl_2O_3$  prelayer. The texture increased with film thickness. A current density of 5 mA/cm<sup>2</sup>, and a stirred solution of 0.1 M TiNO<sub>3</sub> in 5 M NaOH was used to deposit the prelayers.

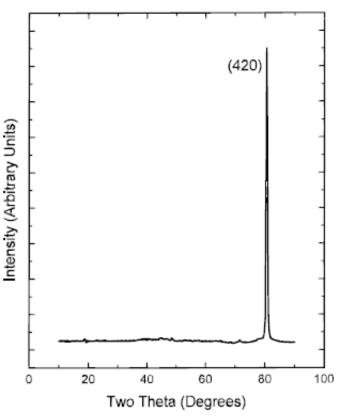


Figure 4. Highly oriented 20  $\mu$ m Pb-T1-O prelayer with a [210] texture. A current density of 5 mA/cm<sup>2</sup>, and a stirred solution of 0.005 M T1NO<sub>3</sub>, 0.1 M Pb(NO<sub>3</sub>)<sub>2</sub> in 5 M NaOH was used to deposit the Pb-T1-O film.



#### XRD can be used for:

- -Bravais lattice determination phase determination (crystalline phases and orientation)
- -Lattice parameter determination
- -Determination of solvus line in phase diagrams (order-disorder transformation)
- -Long range order (Texture Analysis)
- -Crystallite size and Strain
- -Temperature factor thermal diffuse scattering (thermal expansion)

Advanced

-Thickness measurements of thin films and multilayers

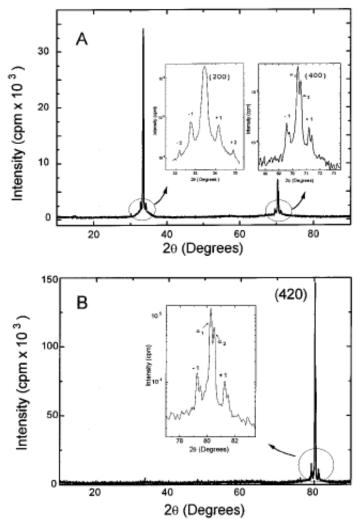


Figure 12. (A) [100]-textured superlattice in the Pb–Tl–O system. The modulation wavelength was calculated as  $\Lambda_F=13.2$  nm by Faraday's law and measured by X-ray diffraction as  $\Lambda_x=13.3$  nm. (B) [210]-textured superlattice with  $\Lambda_F=12$  nm and  $\Lambda_x=11.8$  nm. Both superlattices were deposited by pulsing the current between 0.05 mA/cm² (82 s) and 5 mA/cm² (1.2 s).

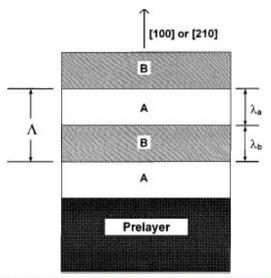


Figure 8. Idealized superlattice grown on an oriented prelayer. The orientation of the prelayer is maintained through the superlattice. The thickness of layer a is  $\lambda_a$ , and the thickness of layer b is  $\lambda_b$ .



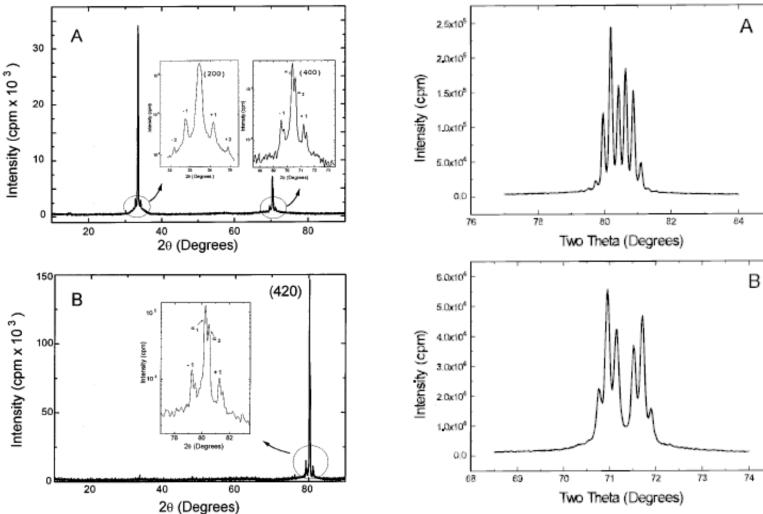


Figure 12. (A) [100]-textured superlattice in the Pb–Tl–O system. The modulation wavelength was calculated as  $\Lambda_{\rm F}=13.2$  nm by Faraday's law and measured by X-ray diffraction as  $\Lambda_x=13.3$  nm. (B) [210]-textured superlattice with  $\Lambda_{\rm F}=12$  nm and  $\Lambda_x=11.8$  nm. Both superlattices were deposited by pulsing the current between 0.05 mA/cm² (82 s) and 5 mA/cm² (1.2 s).

Figure 14. (A) X-ray diffraction pattern of a [210]-textured superlattice in the Pb–Tl–O system using Cu Kα radiation as the X-ray source. The modulation wavelength was calculated as  $\Lambda_F = 50$  nm by Faraday's law. (B) Same as (A), except using Cu K $\beta$  radiation for the X-ray source.

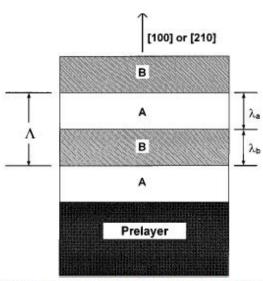


Figure 8. Idealized superlattice grown on an oriented prelayer. The orientation of the prelayer is maintained through the superlattice. The thickness of layer a is  $\lambda_a$ , and the thickness of layer b is  $\lambda_b$ .

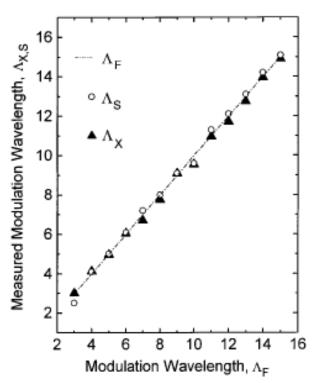


Figure 13. Comparison of the Pb–Tl–O superlattice modulation wavelength in nanometers calculated by Faraday's law  $(\Lambda_F)$  with those measured by scanning tunneling microscopy  $(\Lambda_S)$  and X-ray diffraction  $(\Lambda_X)$ .



# Homework Assignment: Phase Diagram Calculations Due Tuesday 11-19-24

Read Chapter 8 from:

-X-ray Diffraction Procedures by Klug and Alexander

Read Chapter 3-7, 9-11 and 13 from:

-Introduction to X-ray powder
Diffractometry by Jenkins and Synder

Read Chapter 3, 4, 6, 13, and 14 from

-Elements of X-ray Diffraction

by Cullity and Stock

**Read Chapter 2 from Norton**